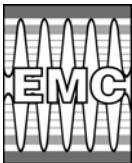

High Doping Effects on In-situ and Ex-situ Ohmic Contacts to n-InGaAs

Ashish Baraskar*, Mark A. Wistey, Vibhor Jain, Uttam Singisetti, Greg Burek,
Brian J. Thibeault, Arthur C. Gossard and Mark J. W. Rodwell
ECE and Materials Departments, University of California, Santa Barbara

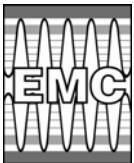
Yong J. Lee

Intel Corporation, Technology Manufacturing Group, Santa Clara, CA



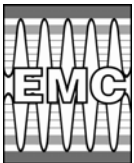
Outline

- **Motivation**
 - Low resistance contacts for high speed HBTs
 - Approach
- **Experimental details**
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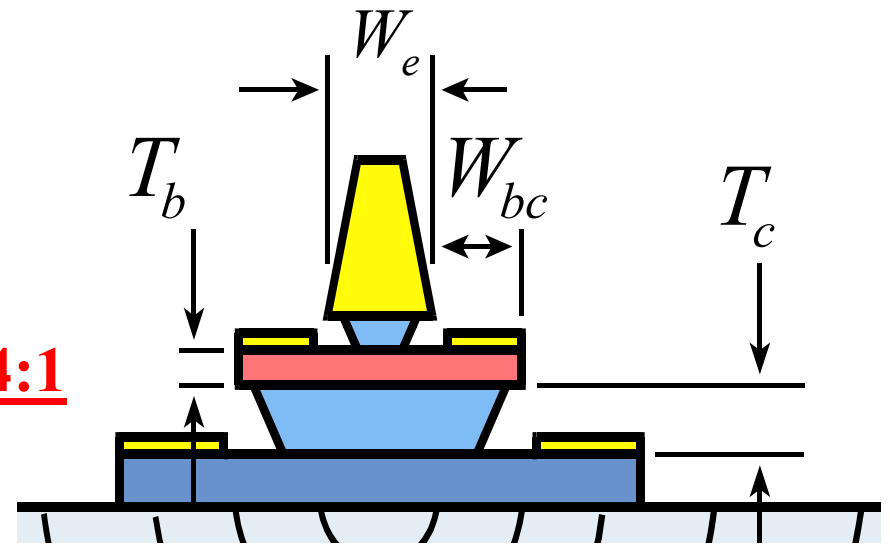
Device Bandwidth Scaling Laws for HBT

To double device bandwidth:

- Cut transit time 2x:
 - Reduce thickness 2:1 ☺
 - Capacitance increases 2:1 ☹
- Cut RC delay 2x
 - **Scale contact resistivities by 4:1**

$$\frac{1}{2\pi f_{\tau}} = \tau_{in} + RC$$

$$f_{\max} = \sqrt{\frac{f_{\tau}}{8 \cdot \pi \cdot (R_{bb} \cdot C_{cb})_{\text{eff}}}}$$



HBT: Heterojunction Bipolar Transistor

Uttam Singiseti, DRC 2007

*M.J.W. Rodwell, IEEE Trans. Electron. Dev., 2001



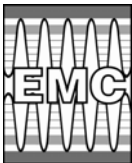
InP Bipolar Transistor Scaling Roadmap

Emitter:	512	256	128	64	32	width (nm)
	16	8	4	2	1	access ρ , ($\Omega \cdot \mu\text{m}^2$)
Base:	300	175	120	60	30	contact width (nm)
	20	10	5	2.5	1.25	contact ρ ($\Omega \cdot \mu\text{m}^2$)
f_t :	370	520	730	1000	1400	GHz
f_{max} :	490	850	1300	2000	2800	GHz

- Contact resistance serious barrier to THz technology

Less than 2 $\Omega \cdot \mu\text{m}^2$ contact resistivity required for simultaneous THz f_t and f_{max} *

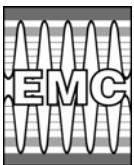
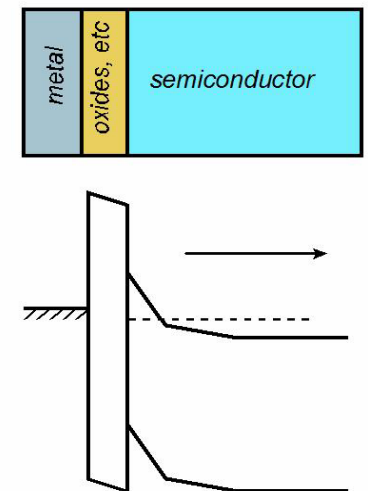
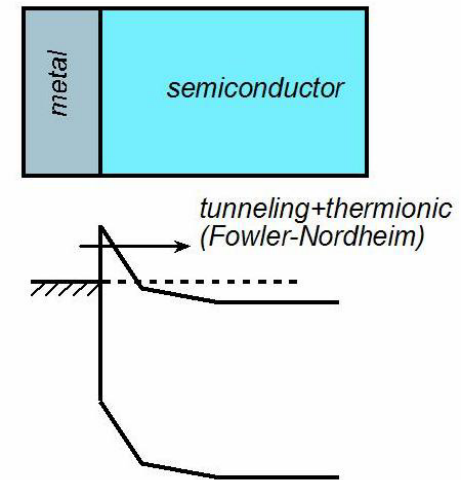
*M.J.W. Rodwell, CSICS 2008



Approach

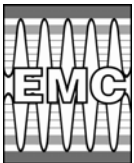
To achieve low resistance, stable ohmic contacts

- **Higher number of active carriers**
 - Reduced depletion width
 - Enhanced tunneling across metal-semiconductor interface
- **Better surface preparation techniques**
 - Ex-situ contacts: treatment with UV-O₃, HCl etch
 - In-situ contacts: no air exposure before metal deposition
- **Use of refractory metal for thermal stability**



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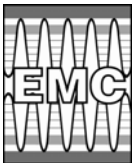


Epilayer Growth

Semiconductor epilayer growth by Solid Source Molecular Beam Epitaxy (SS-MBE)– n-InGaAs/InAlAs

- Semi insulating InP (100) substrate
- Unintentionally doped InAlAs buffer
- Electron concentration determined by Hall measurements

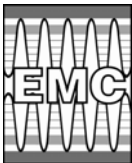
100 nm $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$: Si (n-type)
150 nm $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$: NID buffer
Semi-insulating InP Substrate



Two Types of Contacts Investigated

- In-situ contacts: **Mo**
 - Samples transferred under vacuum for contact metal deposition
 - no air exposure
- Ex-situ contacts: **Ti/Ti_{0.1}W_{0.9}**
 - exposed to air
 - surface treatment before contact metal deposition

Contact metal
100 nm In _{0.53} Ga _{0.47} As: Si (n-type)
150 nm In _{0.52} Al _{0.48} As: NID buffer
Semi-insulating InP Substrate



In-situ contacts

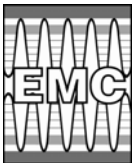
In-situ Molybdenum (Mo) deposition

- E-beam chamber connected to MBE chamber

Why Mo?

- Refractory metal (melting point ~ 2623 C)
- Work function $\sim 4.6 (\pm 0.15)$ eV, close to the conduction band edge of InGaAs
- Easy to deposit by e-beam technique
- Easy to process and integrate in HBT process flow

20 nm in-situ Mo
100 nm $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$: Si (n-type)
150 nm $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$: NID buffer
Semi-insulating InP Substrate

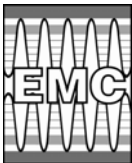
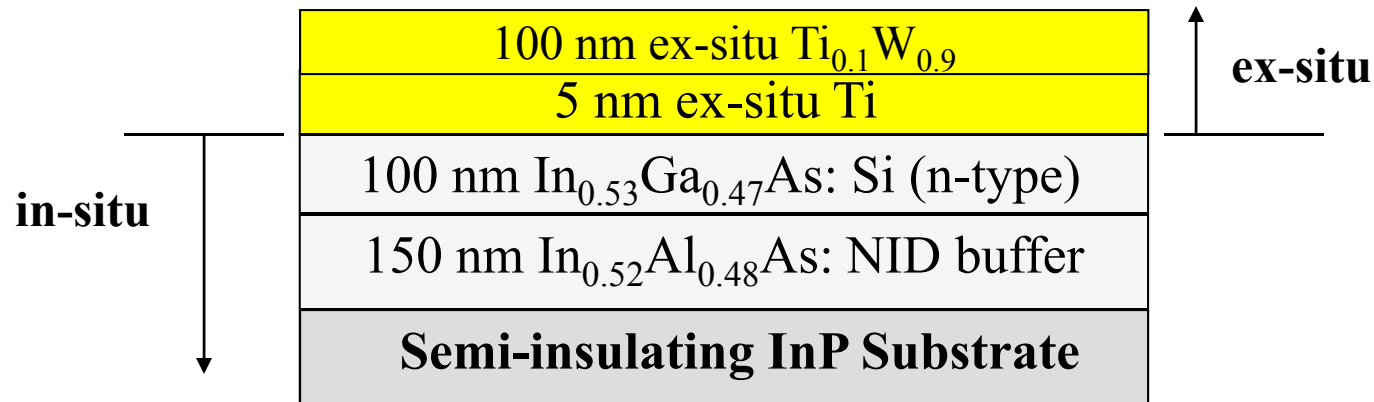


Ex-situ contacts

Ex-situ $\text{Ti}/\text{Ti}_{0.1}\text{W}_{0.9}$ contacts on InGaAs

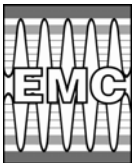
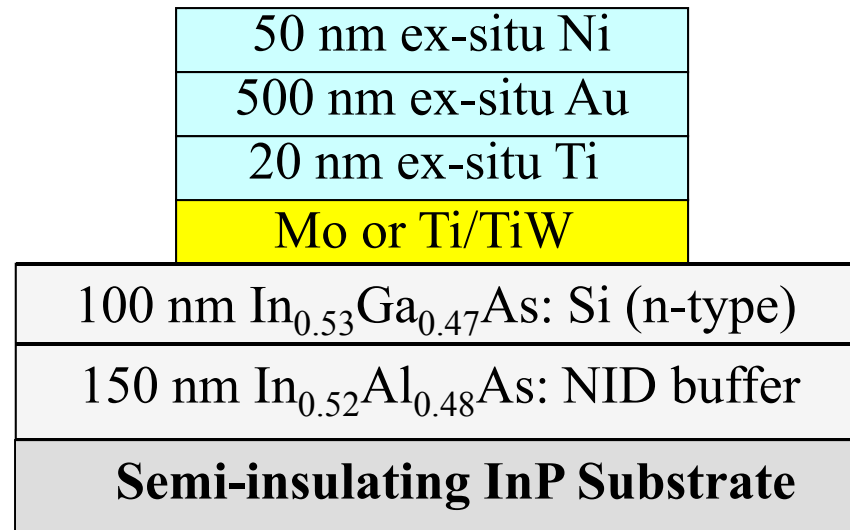
- Surface preparation
 - Oxidized with UV-ozone for 10 min
 - Dilute HCl (1:10) etch and DI rinse for 1 min each
- Immediate transfer to sputter unit for contact metal deposition
- Ti: Oxygen gettering property, forms good ohmic contacts*

*G. Stareev, H. Künzel, and G. Dortmann, *J. Appl. Phys.*, 74, 7344 (1993).



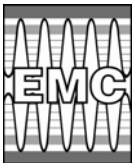
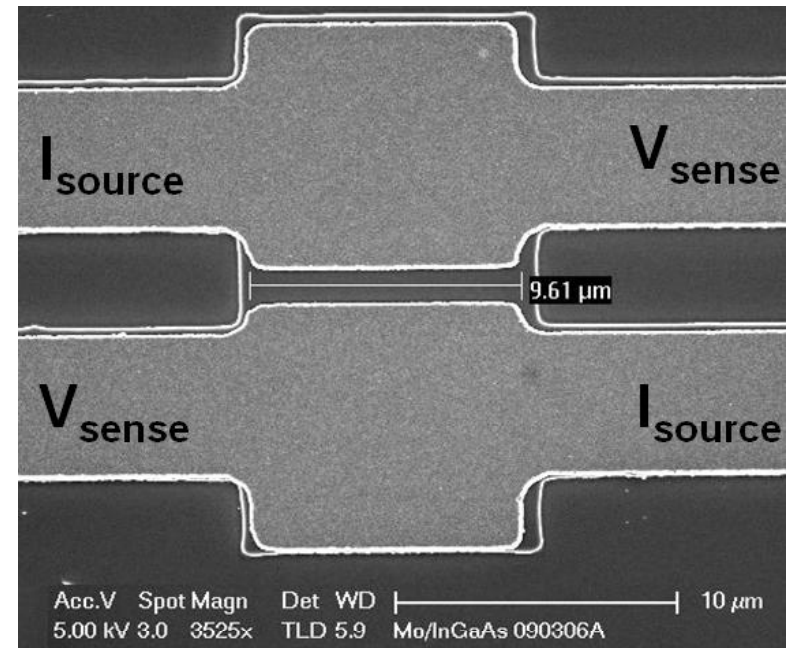
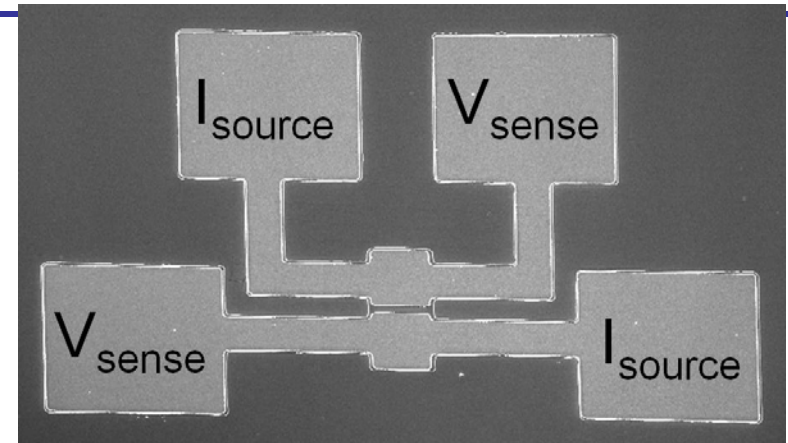
TLM (Transmission Line Model) fabrication

- E-beam deposition of Ti, Au and Ni layers
- Samples processed into TLM structures by photolithography and liftoff
- Mo and Ti/TiW dry etched in SF₆/Ar with Ni as etch mask, isolated by wet etch



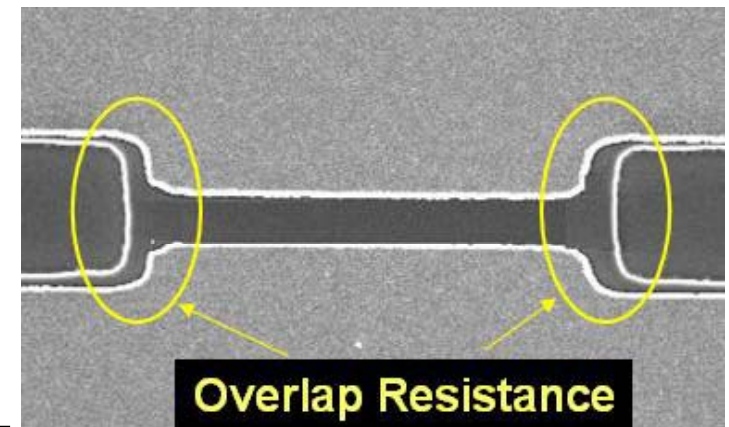
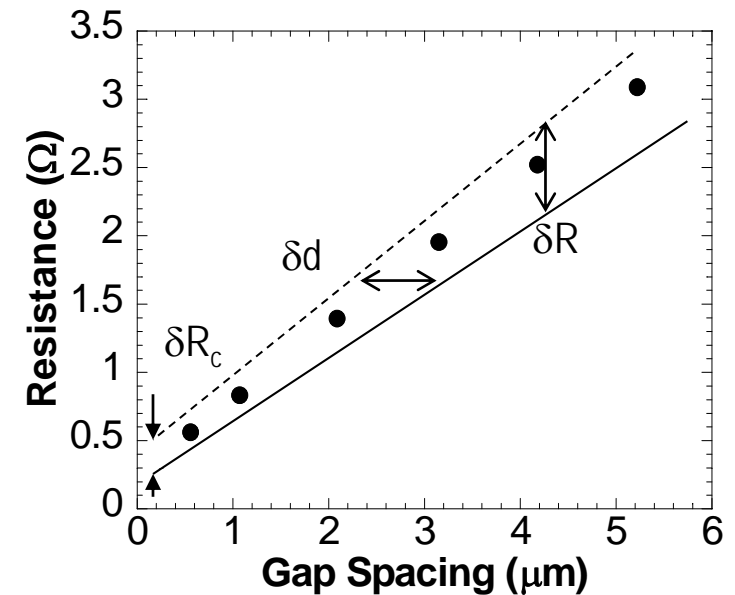
Resistance Measurement

- Resistance measured by Agilent 4155C semiconductor parameter analyzer
- TLM pad spacing varied from $0.6\text{-}26\ \mu\text{m}$; verified from scanning electron microscope
- TLM Width $\sim 10\ \mu\text{m}$

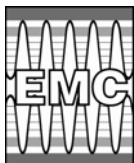


Error Analysis

- Error due to extrapolation*
 - 4-point probe resistance measurements on Agilent 4155C
 - For the smallest TLM gap, R_c is 40% of total measured resistance
- Metal Resistance
 - Minimized using thick metal stack
 - Minimized using small contact widths
 - Correction included in data
- Overlap Resistance
 - Higher for small contact widths

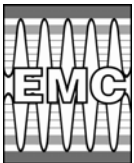


*Haw-Jye Ueng, IEEE TED 2001

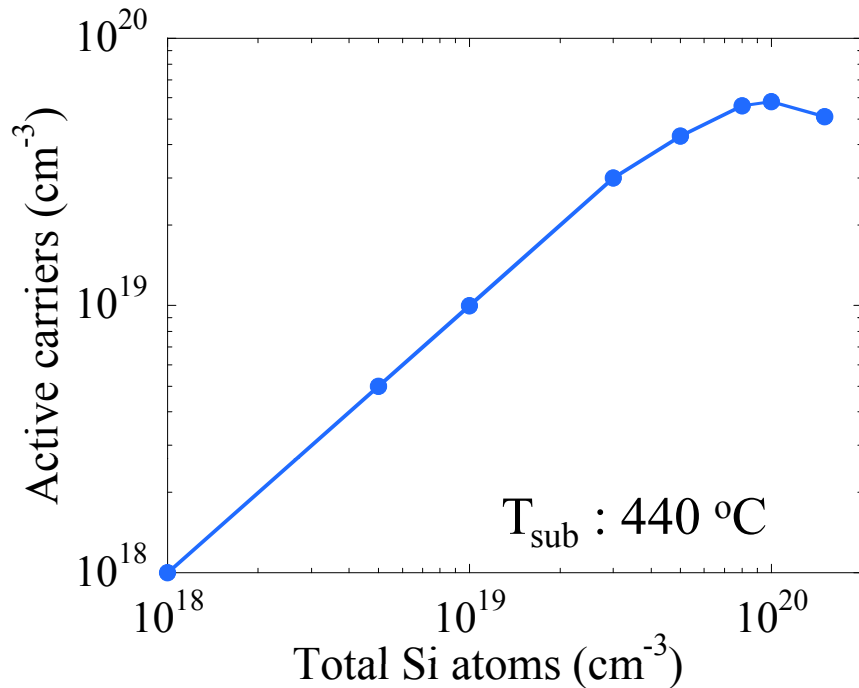


Outline

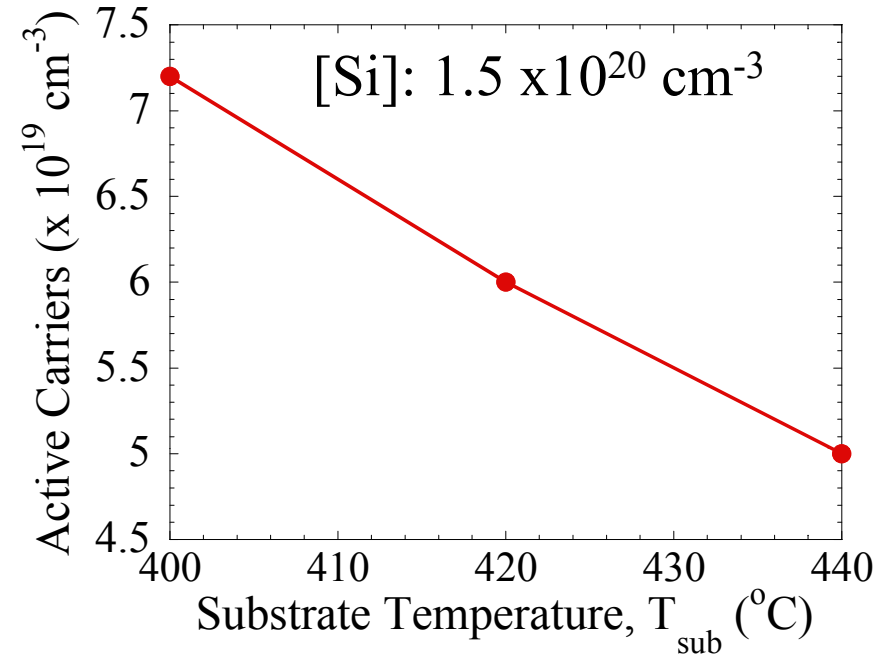
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Results: Doping Characteristics

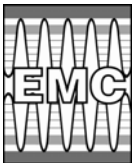


n saturates at high dopant concentration



Enhanced n for colder growths

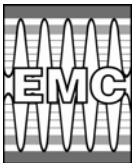
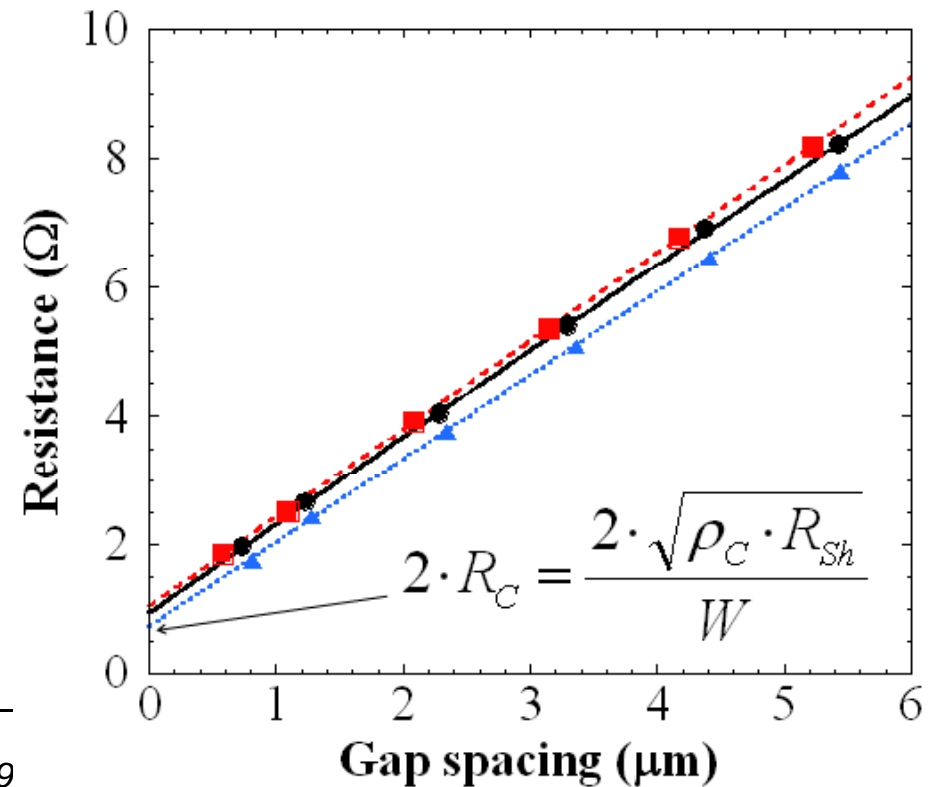
-hypothesis: As-rich surface drives Si onto group-III sites



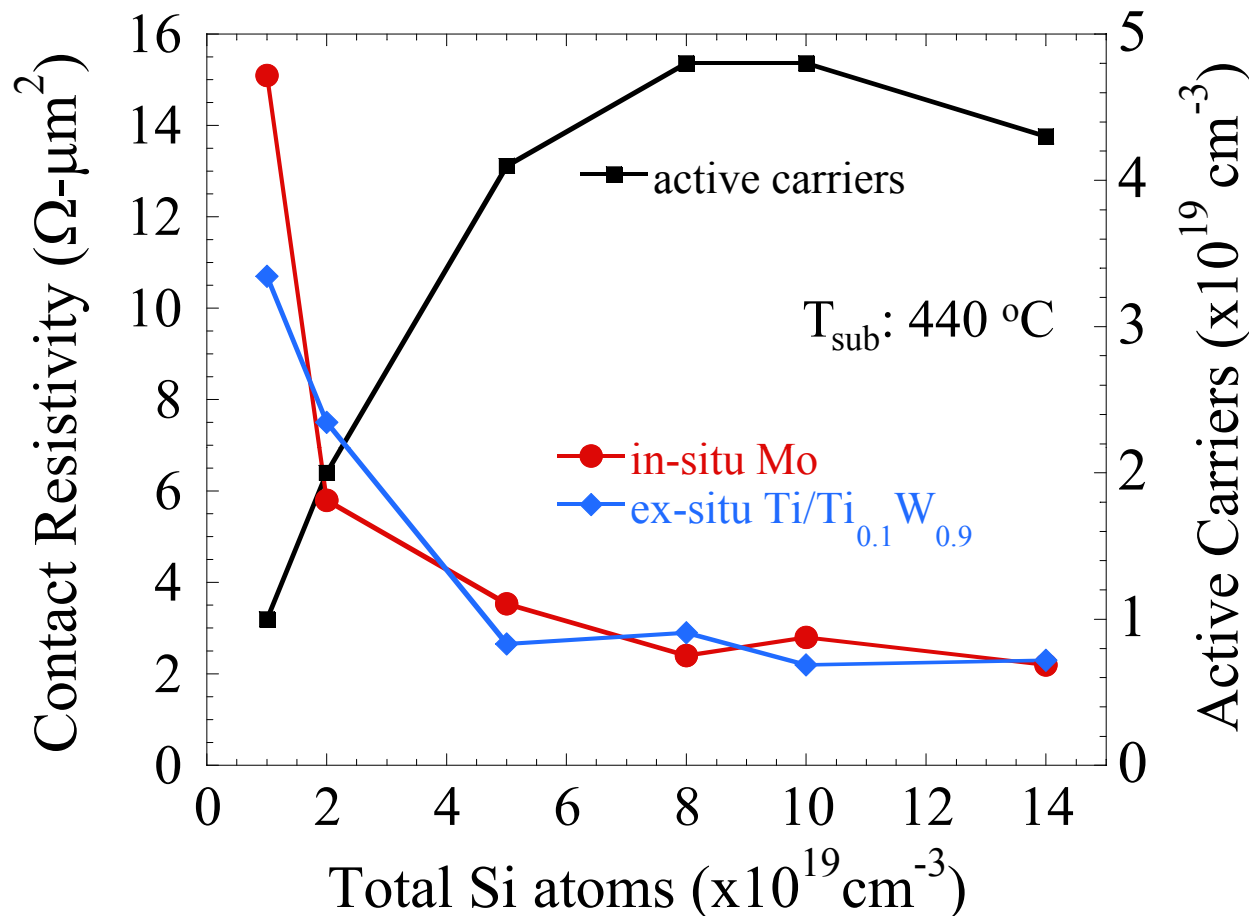
Results: Contact Resistivity

Metal Contact	Active Carriers (cm ⁻³)	ρ_c ($\Omega\text{-}\mu\text{m}^2$)
In-situ Mo	6×10^{19}	1.1 ± 0.6
In-situ Mo	4.2×10^{19}	2.0 ± 1.1
Ex-situ Ti/Ti _{0.1} W _{0.9}	4.2×10^{19}	2.1 ± 1.2

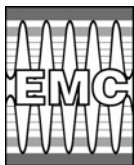
- Mo contacts: in-situ deposition; clean interface
- Ti: oxygen gettering property



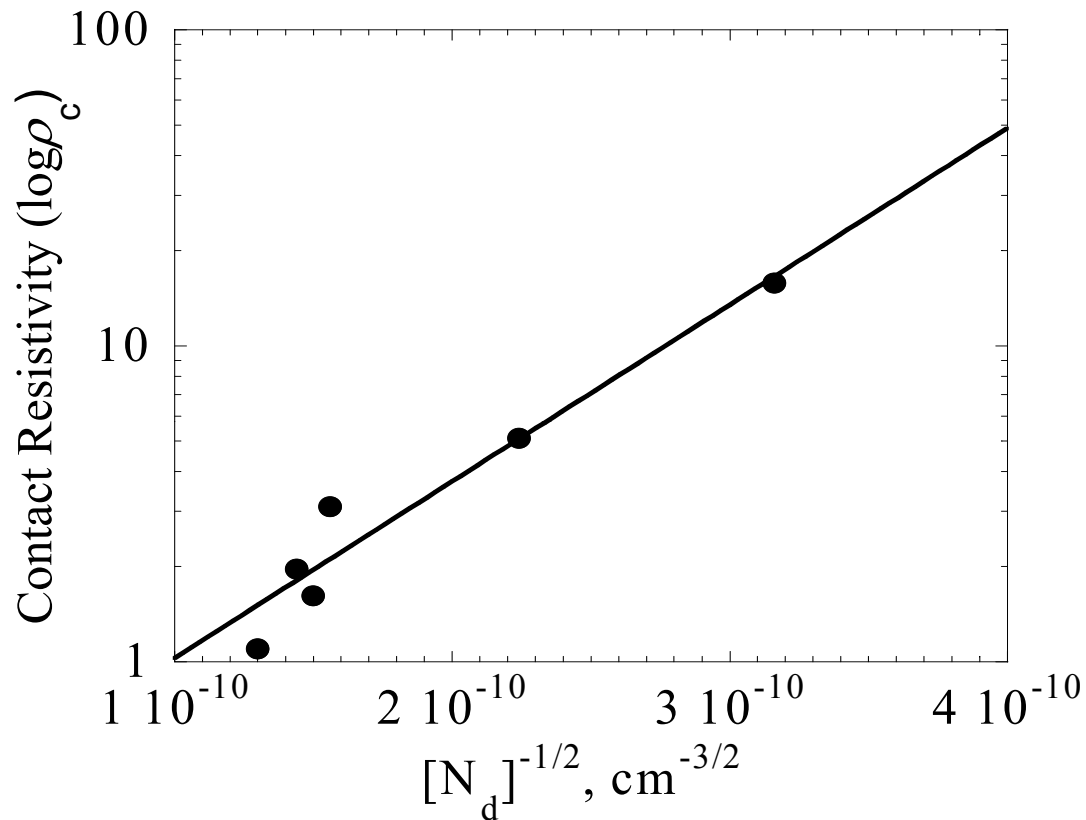
Results: Effect of doping-I



- Contact resistivity (ρ_c) \downarrow with \uparrow in electron concentration



Results: Effect of doping-II



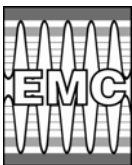
Tunneling $\rightarrow \rho_c \propto \exp\left(\frac{1}{\sqrt{N_d}}\right)^*$

Thermionic Emission $\rightarrow \rho_c \sim \text{constant}^*$

Data suggests tunneling.

High active carrier concentration is the key to low resistance contacts

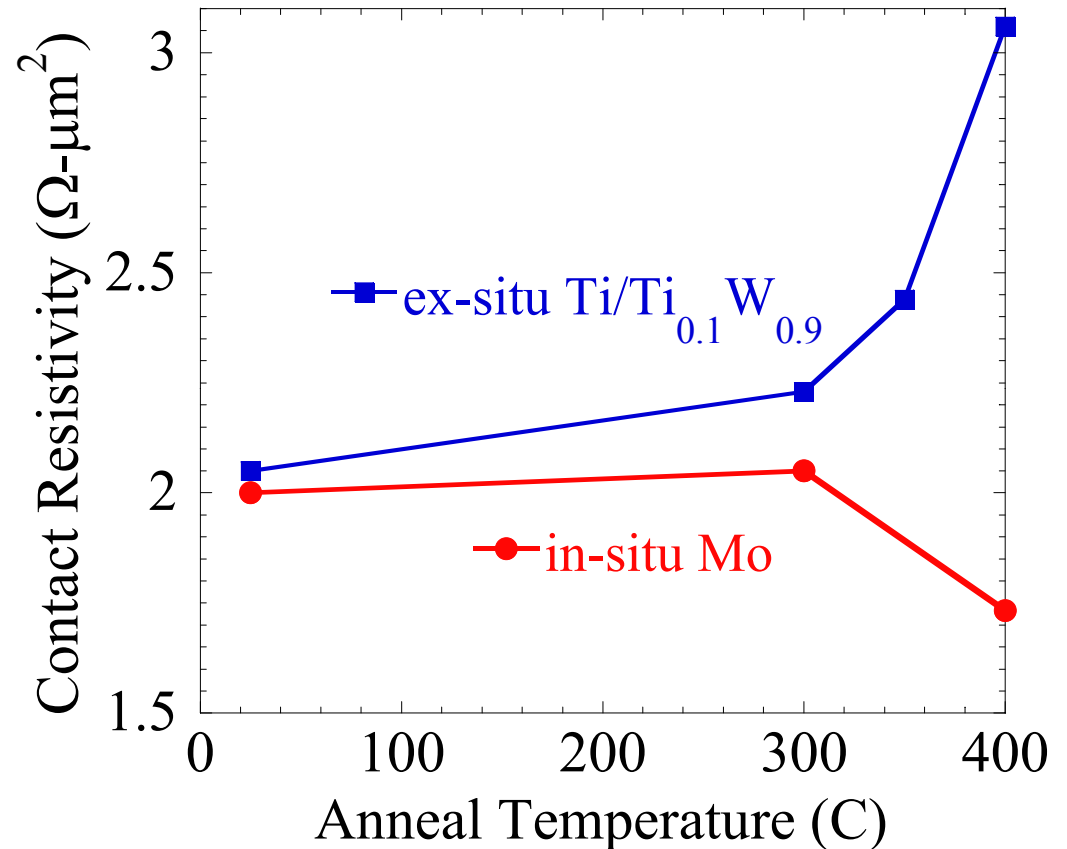
* Physics of Semiconductor Devices, SM Sze



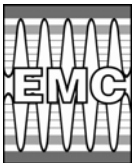
Results: Thermal Stability

Contacts annealed under N_2 flow for 60 secs

- Mo contacts stable to at least 400 C
- $Ti/Ti_{0.1}W_{0.9}$ contacts degrade on annealing*



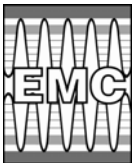
*T. Nittono, H. Ito, O. Nakajima, and T. Ishibashi, *Jpn. J. Appl. Phys., Part 1* **27**, 1718 (1988).



Conclusion

- Extreme Si doping improves contact resistance
- In-situ Mo and ex-situ Ti/Ti_{0.1}W_{0.9} give low contact resistance
 - Mo contacts are thermally stable
 - Ti/Ti_{0.1}W_{0.9} contacts degrade
- $\rho_c \sim (1.1 \pm 0.6) \Omega\text{-}\mu\text{m}^2$ for in-situ Mo contacts
 - less than $2 \Omega\text{-}\mu\text{m}^2$ required for simultaneous THz f_t and f_{max}

✓ **Contacts suitable for THz transistors**

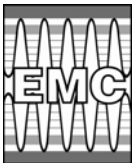


Thank You !

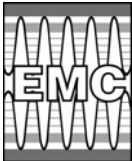
Questions?

Acknowledgements

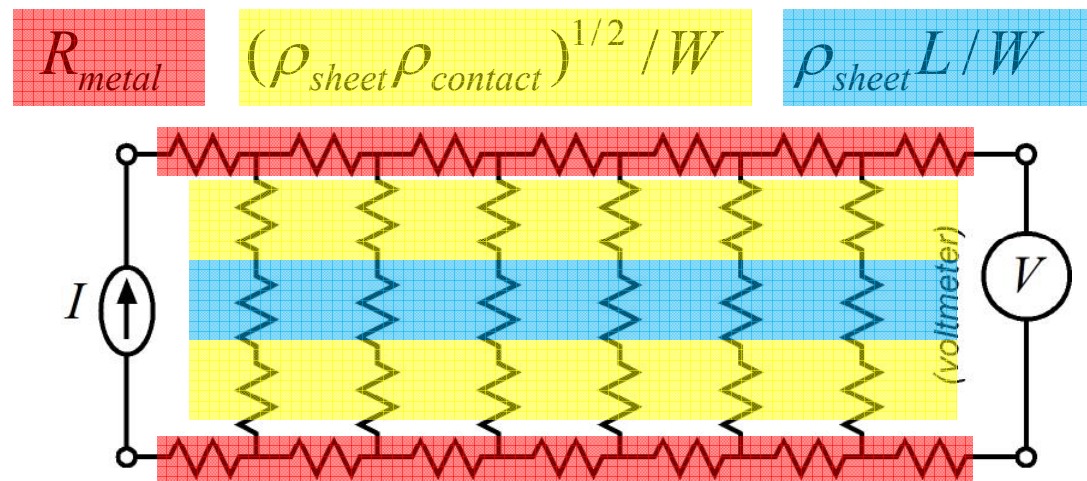
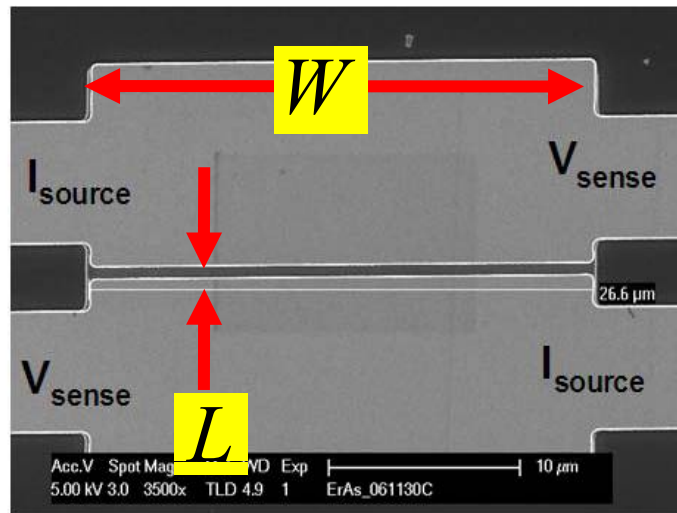
ONR, DARPA-TFAST, DARPA-FLARE



Extra Slides



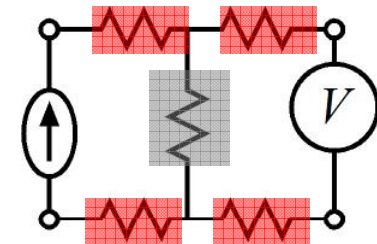
Correction for Metal Resistance in 4-Point Test Structure



$$(\rho_{sheet} \rho_{contact})^{1/2} / W + \rho_{sheet} L / W - R_{metal} / 3$$

From hand analysis & finite element simulation

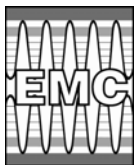
$$R_{metal} / 2$$



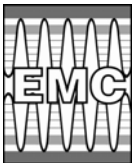
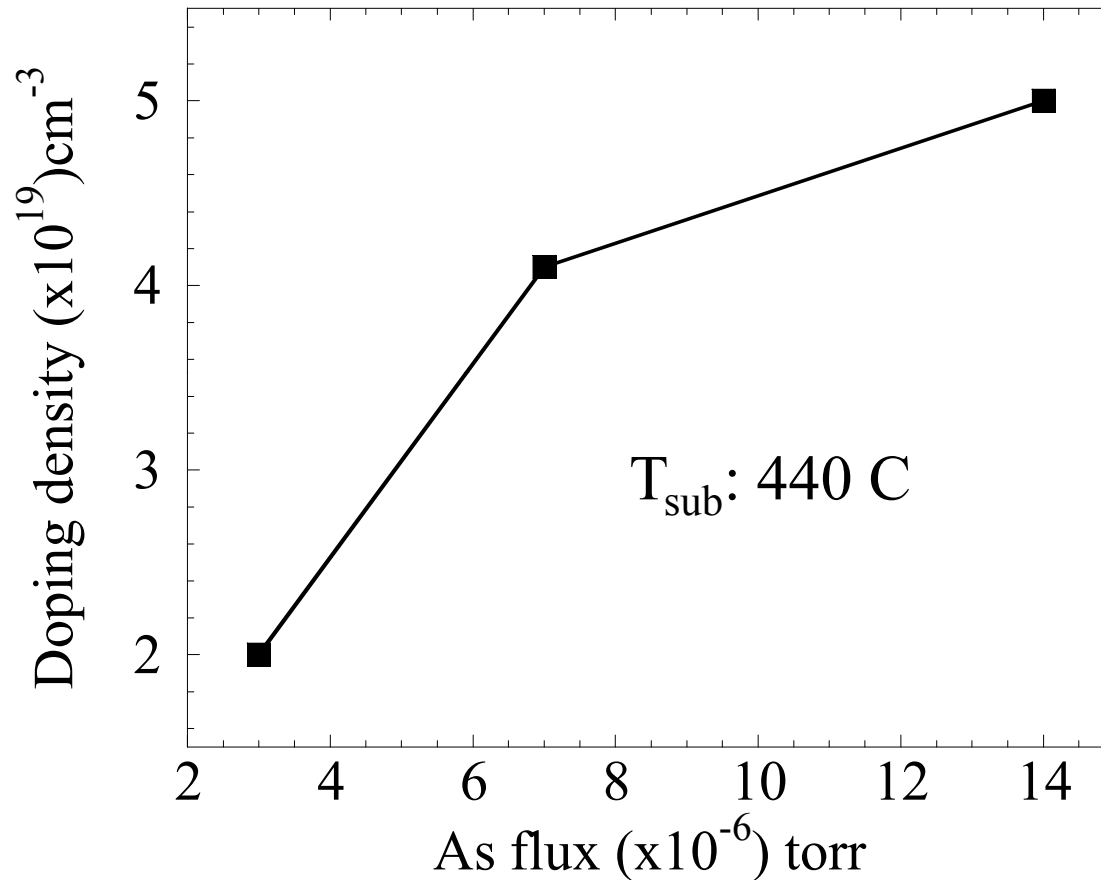
Error term $(-R_{metal}/3)$ from metal resistance

Effect changes measured ρ_c by $\sim 40\%$ ($@ 1.3 \Omega\text{-}\mu\text{m}^2$)

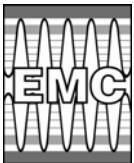
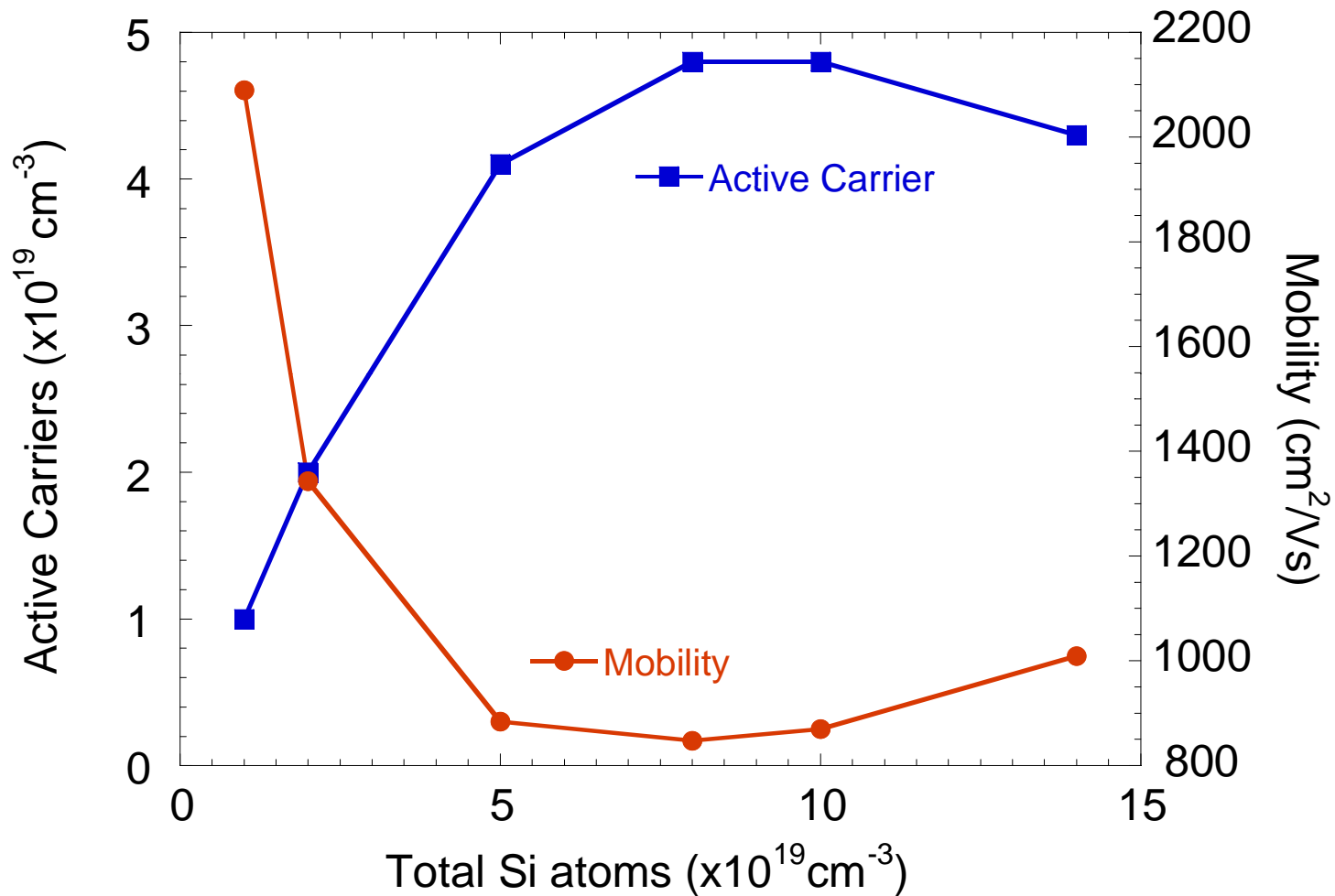
All data presented corrects for this effect



Doping Vs As flux



Active Carrier, Mobility Vs Total Si

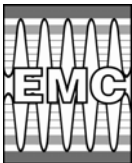


Strain Effects

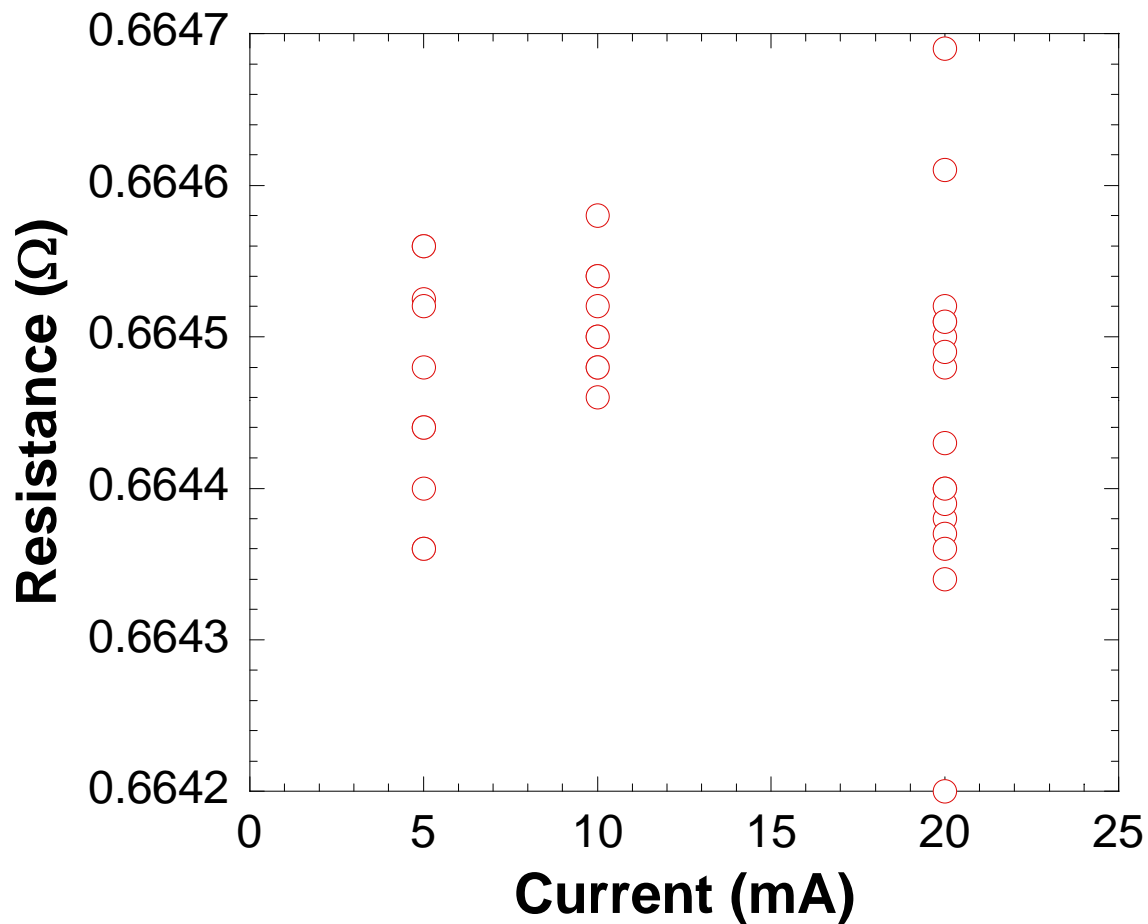
$$[\text{Si}] = 1.5 \times 10^{20} \text{ cm}^{-3}, n = 6 \times 10^{19} \text{ cm}^{-3}$$

$$(a_{\text{sub}} - a_{\text{epi}})/a_{\text{sub}} = 5.1 \times 10^{-4}$$

Van de Walle, C. G., Phys. Rev. B **39**, 3 (1989) 1871



Random and Offset Error in 4155C



- **Random Error in resistance measurement ~ 0.5 mΩ**
- **Offset Error < 5 mΩ***

*4155C datasheet



Accuracy Limits

- **Error Calculations**
 - **dR = 50 mΩ (Safe estimate)**
 - **dW = 1 μm**
 - **dGap = 20 nm**
- **Error in $\rho_c \sim 40\%$ at $1.1 \Omega\text{-}\mu\text{m}^2$**

