

Title: Co-Doping of $\text{In}_x\text{Ga}_{1-x}\text{As}$ with Silicon and Tellurium for Ultra-Low Contact Resistance

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The high dopant concentrations achievable by MBE provide a method for creating low-resistance ohmic contacts: increased doping decreases the depletion depth and increases tunneling.¹ Co-doping of the InAs (and thus $\text{In}_x\text{Ga}_{1-x}\text{As}$) system with a group IV (Sn) and a group VI (Te) is possible and can provide high doping densities ($2.9 \times 10^{19} \text{ cm}^{-3}$).² As the group IV element must sit on a group III site to be a donor and the group VI element sits on the group V site, there is limited competition between the group IV and VI for a specific lattice site, which should allow for increased active electrical carriers. Additionally, Te has a lower diffusivity than Si (in GaAs), which will prevent unwanted movement of dopant atoms during thermal cycling.³ We show that Te can dope $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ up to $2.6 \times 10^{19} \text{ cm}^{-3}$, and that the contact resistivity between an ex-situ deposited metal and Si and Te co-doped InAs and $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ is lower than that of InAs and $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ doped only by Si.

Samples were grown by solid source MBE lattice matched to semi-insulating InP with layer structure as follows from the substrate. Hall samples: 150 nm $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$, 500 nm Te doped $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ with Te (GaTe source material) cell temperatures of 475, 487, 500, 520, 525, 550, 562, 582, and 625 °C. Sample A: 400 nm $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$, 3 nm of Si-doped $1 \times 10^{19} \text{ cm}^{-3}$ $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$, 7 nm of $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$, 60 nm (Si or Si and Te doped) $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$. Sample B: 400 nm $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$, 3 nm of Si-doped $2 \times 10^{19} \text{ cm}^{-3}$ $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$, 10 nm of $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$, and 60 nm (Si or Si and Te doped) InAs. Samples A and B were removed from vacuum prior to growth of the top n^+ layer. To clean the surface prior to reloading to vacuum samples were oxidized with UV ozone and those oxides were removed by a 1 min. dip in 10 H₂O:1 HCL. Samples were heated to 420 °C and treated with thermally cracked hydrogen ($\approx 1 \times 10^{-6}$ Torr) for 40 minutes prior to regrowth. After regrowth Samples A and B were metalized with lifted-off e-beam evaporated Ti/Pd/Au and mesa isolated. Contact resistances were extracted by transmission line measurements (TLM).

Room temperature Hall measurement of the Te doped $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ Hall samples using standard Van der Pauw technique showed electrically active carrier concentrations of up to $2.6 \times 10^{19} \text{ cm}^{-3}$ (with mobility of $1466 \text{ cm}^2/\text{Vs}$) without saturation effects. The contact resistivity to the Si and Te co-doped InAs (sheet resistance of $18.9 \ \Omega$) was $6.6 \ \Omega\text{-}\mu\text{m}$ ($2.3 \ \Omega\text{-}\mu\text{m}^2$) while the contact resistivity to the Si doped InAs (sheet resistance of $25.3 \ \Omega$) was $9.9 \ \Omega\text{-}\mu\text{m}$ ($3.9 \ \Omega\text{-}\mu\text{m}^2$). The contact resistivity to the Si and Te co-doped $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ (sheet resistance of $24 \ \Omega$) was $6.8 \ \Omega\text{-}\mu\text{m}$ ($1.9 \ \Omega\text{-}\mu\text{m}^2$) while the contact resistivity to the Si doped $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ (sheet resistance of $31 \ \Omega$) was $8.5 \ \Omega\text{-}\mu\text{m}$ ($2.3 \ \Omega\text{-}\mu\text{m}^2$). These results suggest that co-doping of $\text{In}_x\text{Ga}_{1-x}\text{GaA}$ by Si and Te is an effective way to lower the metal-semiconductor contact resistance and resistivity. Hall and AFM measurements of co-doped are forthcoming.

¹ A. K. Baraskar *et al.*, JVST B **27**, 2036 (2009).

² H. G. Lee *et al.*, J. Cryst. Growth **130**, 416 (1993).

³ R. Sankaran, J. Cryst. Growth **50**, 859 (1980).

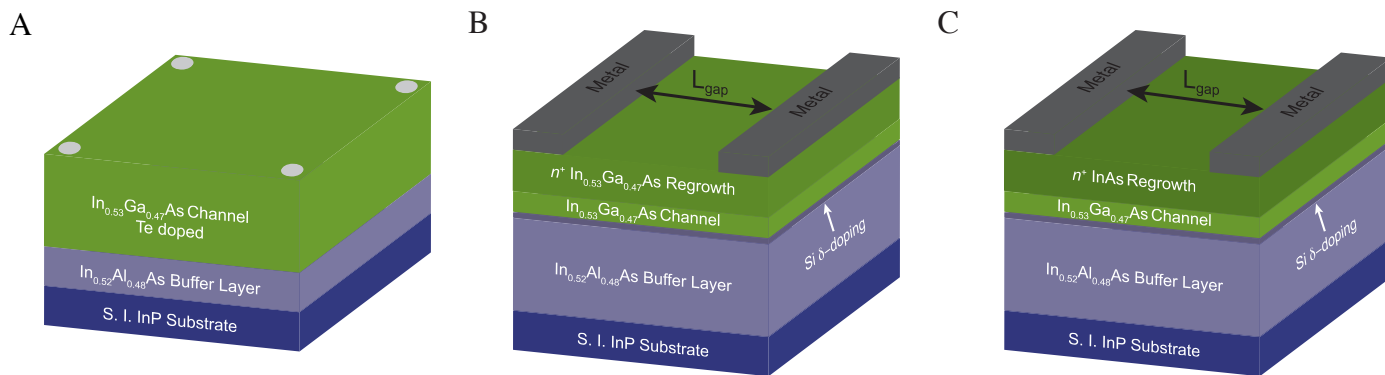


Fig. 1: Schematic illustration of the sample structures (a) Hall sample (b) Sample B (v7 nm channel with $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ regrowth) and (c) Sample C (10 nm channel with InAs regrowth)

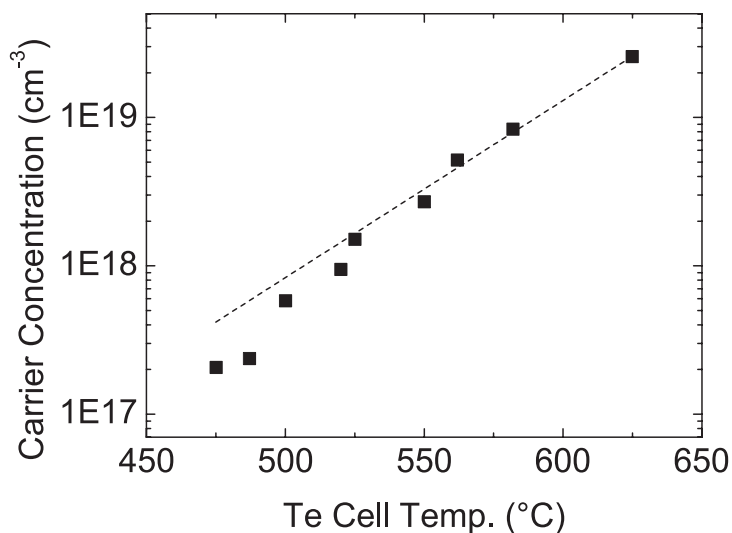


Fig. 2: Plot of Te cell temperature versus measured free electron concentration as measured by Hall.

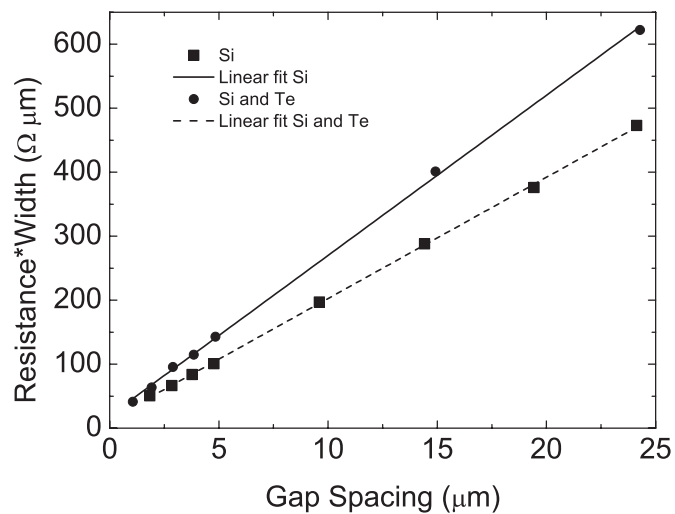


Fig. 3: Plots of measured resistance versus gap spacing for Si-doped InAs regrowth (circles) and Si- and Te- doped InAs regrowth (squares) with linear fits as solid and dashed line, respectively.

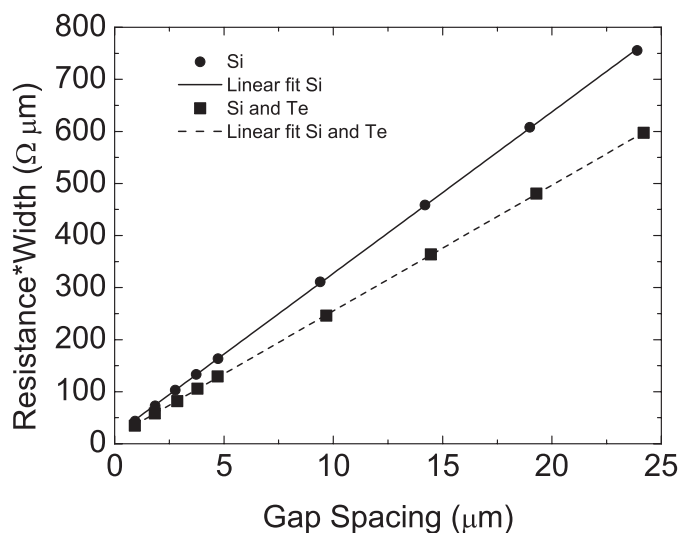


Fig. 4: Plots of measured resistance versus gap spacing for Si-doped InGaAs regrowth (circles) and Si- and Te- doped InGaAs regrowth (squares) with linear fits as solid and dashed line, respectively.